The Crystal Structure of Hf$_3$Cu$_8$ and Zr$_3$Cu$_8$

By Lars Bsenko

Institute of Chemistry, University of Uppsala, Box 531, S-751 21 Uppsala, Sweden

The crystal structure of the intermetallic phase Hf$_3$Cu$_8$ has been determined from three-dimensional single-crystal X-ray diffractometer data. The space group is *Pnma*, and the unit cell, of dimensions $a = 7.8113$ (5), $b = 8.1014$ (6), $c = 9.9470$ (7) Å, $V = 629.60$ Å^3, contains four formula units. Full-matrix least-squares refinement gave a conventional $R(F)$ value of 0.053. Two non-equivalent sets of Hf atoms occur in the structure, one with a coordination number of 13 and the other of 14. The six non-equivalent sets of Cu atoms have coordination numbers ranging from 11 to 13. X-ray powder diffraction data show that Zr$_3$Cu$_8$ is isostructural with Hf$_3$Cu$_8$.

Introduction

The first observation of a peritectic phase at about 70 at. % Cu in the Zr–Cu system was made by Lundin, McPherson & Hansen (1953). The phase was tentatively assigned the formula Zr$_2$Cu$_5$ and was thought to be formed by a reaction between the melt and a phase of composition ZrCu$_3$ or ZrCu$_4$. Using electron-probe microanalysis, Donachie (1963–1964) stated that the composition of the congruently melting phase is ZrCu$_4$. Perry & Hugi (1972) also found a phase at the composition ZrCu$_4$ and reported the formation of the peritectic phase Zr$_2$Cu$_5$. Later, Gabathuler, White & Parthé (1975) and Bsenko (1975), independently found that the phase denoted ZrCu$_4$ possessed the GdAg$_{3.6}$ type of structure determined by Bailey & Kline (1971), and hence had the ideal composition Zr$_{14}$Cu$_{51}$. Bsenko (1975) showed, by indexing the powder pattern, that the peritectic phase is orthorhombic. The Hf–Cu system was investigated by Markiv, Petkov, Storozhenko, Ivanchenko & Gorskii (1974), who found two intermediate phases in the region 70–75 at. % Cu, denoted Hf$_2$Cu$_5$ and HfCu$_3$. Hf$_2$Cu$_5$ was formed at 990°C by the reaction: liq.+HfCu$_3$ = Hf$_2$Cu$_5$. Perry (1974) found a phase denoted HfCu$_4$, which was later shown to be isostructural with GdAg$_{3.6}$ (Gabathuler, White & Parthé, 1975; Bsenko, 1975). It is probable that the phase denoted HfCu$_3$ by Markiv *et al.* is identical with Hf$_{14}$Cu$_{51}$. A phase with orthorhombic symmetry and forming peritectically was found by Bsenko (1975). The present investigation was undertaken in order to establish the structure of this peritectic phase.

Experimental

By arc-melting turnings of zirconium and hafnium metal (Koch-Light, Colnbrook, England; Zr of purity 99·9% containing less than 0·03% oxygen and zone-refined Hf containing 3% Zr) with copper (HCOKOF, certified grade copper, min. 99·99%), two alloys with the nominal compositions Hf$_{0.28}$Cu$_{0.72}$ and Zr$_{0.28}$Cu$_{0.72}$

were prepared. The melting was carried out on a water-cooled copper hearth under a purified argon atmosphere, with a non-consumable tungsten electrode. The alloy buttons were turned upside down and remelted to insure homogeneity. The weight loss was less than 0·5%.

The samples were wrapped in tantalum foil and annealed in sealed and evacuated silica tubes for 500 h at 850°C. After quenching in water the surfaces of the samples were wet-ground on a carborundum disc and polished with alumina. Etching with a mixture consisting of 10 parts of HF, 20 parts of HNO$_3$, and 70 parts of H$_2$O by volume, revealed the microstructure of the surface. Metallographic examination showed that the alloys were nearly single-phase.

The samples were crushed and small fragments were examined in a Weissenberg camera. All fragments turned out to be polycrystalline.

Assuming that the peritectically formed phase contained about 72 at.% copper, an alloy of nominal composition Hf$_{0.30}$Cu$_{0.70}$ was prepared and heat-treated as described earlier. This specimen was placed in a ZrO$_2$ crucible, inside an evacuated and sealed silica tube. The tube was held at a temperature of 1025°C for 24 h. After quenching in water, a metallographic examination of the polished and etched surface showed primary grown crystals, indicating that the annealing temperature was above the eutectic isotherm.

The alloy was crushed and a small fragment proved to be a single crystal when examined in a Weissenberg camera. The diffraction symmetry and the systematic absences of the reflexions indicated that the space group was *Pnma* or *Pn2$_1$a*. Diffractometric intensity data were collected on a four-circle Stoe–Philips X-ray diffractometer controlled by a PDP8/I computer. The $\omega/2\theta$ scan technique was employed and graphite-monochromatized Mo radiation was used ($\lambda = 0.71069$ Å). The 800, 040 and 006 reflexions were used as control reflexions, and during the data collection these were checked every 20th reflexion scanned. The differences in intensity of these reference reflexions were within the limits of the random statistical fluctuations.

F_o^2 and $\sigma_c(F_o^2)$ values were obtained from the integrated peak intensities by applying corrections for background, Lorentz and polarization effects and absorption, assuming the monochromator to be ideally imperfect. A linear absorption factor of 682 cm^{-1} was applied in computing absorption corrections. The crystal was of irregular shape, and was approximated with six boundary planes to a polyhedron with dimensions of the order of $0.06 \times 0.03 \times 0.01$ mm for the purpose of making absorption corrections. The transmission factors were found to vary from 0.213 to 0.520. Data collection was carried out to sin $\theta/\lambda = 0.587$ Å^{-1}, and

Table 1. *Final structural parameters for* Hf$_3$Cu$_8$

E.s.d.'s are in units of the least significant digit. Positional parameters are $\times 10^5$, thermal $\times 10^4$. Anisotropic thermal parameters are in the form exp $[-(\beta_{11}h^2 + \ldots + 2\beta_{12}hk + \ldots)]$.

		x	y	z	β_{11}	β_{22}	β_{33}	β_{12}	β_{13}	β_{23}
Hf(1)	8(d)	6255 (8)	44594 (7)	14022 (6)	21 (1)	22 (1)	11 (1)	−1 (1)	−1 (1)	−1 (1)
Hf(2)	4(c)	45964 (12)	¼	4670 (9)	25 (1)	29 (1)	12 (1)	0	2 (1)	0
Cu(1)	8(d)	37912 (25)	57601 (23)	16013 (18)	22 (2)	30 (2)	14 (1)	−3 (2)	−2 (1)	2 (1)
Cu(2)	8(d)	19893 (23)	49897 (21)	40644 (18)	26 (2)	19 (2)	16 (1)	1 (1)	−2 (1)	1 (1)
Cu(3)	4(c)	28548 (35)	¼	27349 (25)	24 (4)	32 (3)	8 (2)	0	−1 (1)	0
Cu(4)	4(c)	11655 (35)	¼	90329 (26)	18 (3)	23 (3)	16 (2)	0	−5 (2)	0
Cu(5)	4(c)	36932 (35)	¼	76548 (25)	23 (3)	20 (3)	9 (2)	0	−3 (2)	0
Cu(6)	4(c)	32283 (33)	¼	51676 (25)	16 (3)	20 (3)	9 (2)	0	−1 (2)	0

Table 2. *Observed and calculated X-ray structure factors for* Hf$_3$Cu$_8$

The four columns are, in order: k, l, F_o and F_c. Asterisks mark reflexions not included in the final refinement.

[Table of observed and calculated structure factors — dense multi-column numerical data not legibly transcribable.]

a total number of 1734 independent reflexions were recorded. Of these, 1323 were in excess of $2\sigma_c(F_o^2)$ and were used in the final refinement.

From the Weissenberg photographs preliminary cell dimensions were obtained, and these were used in indexing the X-ray powder patterns for $Hf_{0.28}Cu_{02.7}$

and $Zr_{0.28}Cu_{0.28}$. The X-ray powder photographs were taken with a Guinier–Hägg type focusing camera (Philips XDC 700), using zone-refined silicon ($a = 5.43088$ Å) as internal calibration standard and Cr $K\alpha_1$ radiation ($\lambda = 2.28975$ Å). A least-squares refinement using the local program *CELNE* (Lundgren, 1974)

Table 3. *Comparison of calculated and observed Q values and intensities for* Zr_3Cu_8

$Q = 1/d^2$, d is the interplanar spacing in Å. Calculated intensities less than 1.0 are omitted.

| hkl	Q_c (×10⁵)	Q_o (×10⁵)	I_c	I_o	hkl	Q_c (×10⁵)	Q_o (×10⁵)	I_c	I_o
011	2506	2497	2·1	0·5	133	24165	–	3·0	–
121	8629	8615	2·9	3·9	322	24555	24553	38·9	45·3
022	10022	–	1·6	–	400	25836	25819	3·0	15·2
103	10639	10634	4·7	7·1	105	26681	–	1·9	–
122	11637	–	1·1	–	401	26839	–	1·3	–
221	13473	13466	4·6	4·2	410	27339	27354	4·0	–
004	16043	16038	7·6	10·5	042	28057	–	2·5	–
222	16481	16476	19·1	20·0	115	28184	28190	9·0	11·0
213	16986	16989	15·4 }	} 56·5	411	28342	28345	22·5	22·6
311	17038	17049	15·7 }		233	29009 }	} 29039	2·4 }	} 5·3
104	17657	17658	4·9	4·2	331	29061 }		3·7 }	
302	18543	18542	36·8	41·7	142	29671	29668	14·6	17·7
132	19151 }	} 19145	70·4 }	} 100·0	402	29847	–	2·4	–
114	19160 }		29·6 }		240	30505 }	} 30548	2·2 }	} 4·7
230	19985	19980	45·4	50·5	304	30575 }		1·6 }	
231	20988	20983	7·7	8·0	412	31350	–	2·6	–
223	21494 }	} 21521	34·7 }	} 86·6	241	31508	31515	4·5	3·9
321	21547 }		60·5 }		314	32078	–	1·3	–
024	22054	22049	58·3	62·9	242	34515	–	1·2	–
204	22502	22498	4·0	8·6	143	34685	34698	4·4	4·4
033	22560	–	1·5	–	403	34860	34855	7·1	4·8
303	23557	23561	19·3 }	} 71·9	324	36587	–	1·3	–
124	23669	23678	27·3 }						
232	23996 }		5·8 }						
214	24004 }	} 24008	43·9 }	} 85·7					
040	24046 }		33·0 }						
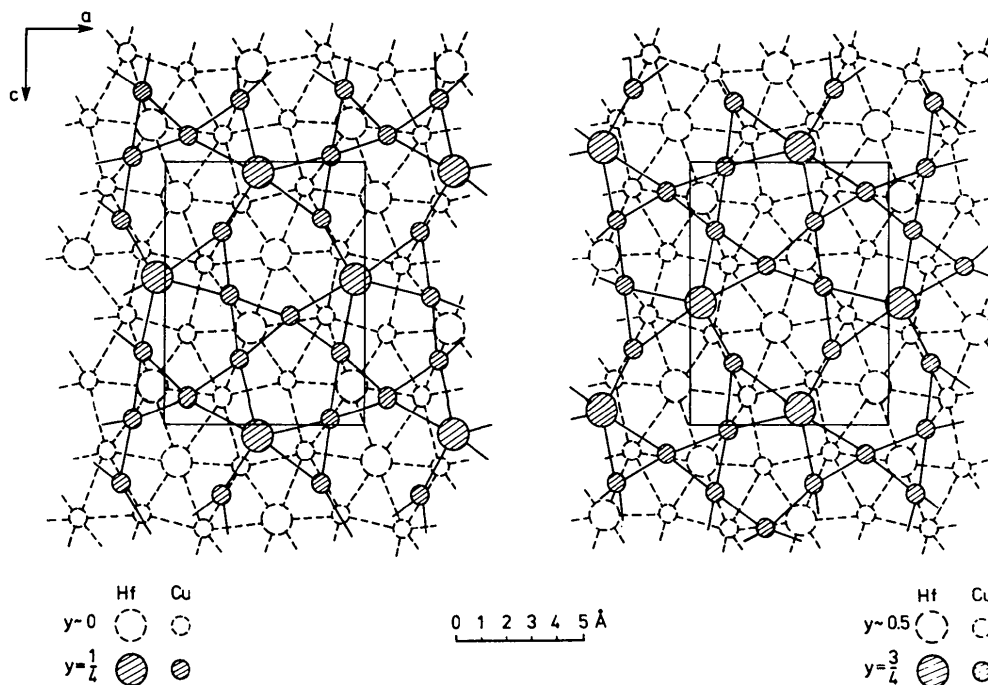
Fig. 1. The two sections of the structure of Hf_3Cu_8 projected on (010).

gave the cell parameters. The X-ray powder photograph of the sample $Zr_{0.28}Cu_{0.72}$ was measured in an automatic film scanner SAAB model 2, and integrated intensities were obtained following the procedure of Malmros & Werner (1973). Some of the reflexions could not be resolved and for these the sums of the integrated intensities were obtained.

Determination and refinement of the Hf₃Cu₈ structure

At the start of the structure determination, the unit-cell content of the crystal was estimated in the following manner. From a plot of X-ray densities against composition, based on crystallographic data for Hf from Russell (1953), for Hf_2Cu from Nevitt & Downey (1962), for Hf_7Cu_{10} and $Hf_{14}Cu_{51}$ from Bsenko (1975) and for Cu from Frohnmeyer & Glocker (1953), theoretical densities were obtained for alloys containing 71–73 at.% Cu. This range should include the composition of the crystal concerned. An examination of the various possibilities yielded a cell content of 12 Hf atoms and 32 Cu atoms as the only alternative compatible with a structure of $Pn2_1a$ or $Pnma$ symmetry without disorder or vacancies. A three-dimensional Patterson synthesis was computed, and the largest maxima could be interpreted in terms of 12 Hf atoms situated in one 8(d) and one 4(c) position in space group $Pnma$. From this starting point, a series of Fourier difference syntheses successively revealed the positions of the Cu atoms.

The full-matrix least-squares program *UPALS* (Lundgren, 1974) was used for refining the structure. The function minimized in the least-squares program was $w(|F_o|-|F_c|)^2$ where $w^{-1}=\sigma_c^2(F_o)+(0.04|F_o|)^2$, with $\sigma_c(F_o)$ based on counting statistics. Atomic scattering factors for Cu were taken from Hanson, Herman, Lea & Skillman (1964) and for Hf from Cromer & Waber (1965). Dispersion correction factors were taken from Cromer & Liberman (1970). No extinction correction was made. The parameters varied were: one scale factor, 19 atomic positional parameters and 38 anisotropic temperature factors. In the last cycle the changes in parameters were all smaller than 0.01σ. The conventional agreement index $R(F)$ was 0·053 for 1323 reflexions with F_o^2 larger than $2\sigma_c(F_o^2)$, and including zero-weight data an $R(F)$ value of 0·076 was obtained for all 1734 reflexions. The positional and the anisotropic thermal parameters are given in Table 1. The anisotropies in the thermal parameters result from the difficulty in applying a proper absorption correction to the irregular crystal. The calculated structure factors are compared with the observed structure factors in Table 2.

The structure of Zr₃Cu₈

With the positional and thermal parameters obtained for Hf₃Cu₈, the powder intensities of the reflexions for Zr₃Cu₈ were calculated. Atomic scattering factors

for Zr were taken from Cromer & Waber (1965). The intensities were obtained with the formula $I_c = g\,Lp\,|F^2|A$ in which g, Lp, F and A represent the multiplicity factor, the Lorentz–polarization factor, the structure factor and the absorption factor for the film respectively. The calculated and observed intensities on a relative scale from 0–100 are given in Table 3. The reliability index $R=\sum|I_o-I_c|/\sum I_c$ is 0·22.

Description of the structure

The structure of Hf₃Cu₈ can be described in terms of layers of atoms parallel to (010) and stacked in a

Table 4. *Interatomic distances shorter than 3·5 Å in* Hf₃Cu₈

E.s.d.'s are in units of the least significant digit.

Hf(1)–Hf(1)	1	3·083 (1) Å	Cu(3)–Hf(1)	2	2·703 (2) Å
Hf(1)	1	3·175 (1)	Hf(1)	2	2·819 (2)
Cu(1)	1	2·666 (2)	Hf(2)	1	2·633 (3)
Cu(1)	1	2·697 (2)	Hf(2)	1	3·112 (3)
Cu(2)	1	2·886 (2)	Cu(1)	2	2·964 (2)
Cu(2)	1	2·910 (2)	Cu(2)	2	2·506 (2)
Cu(2)	1	3·014 (2)	Cu(6)	1	2·438 (3)
Cu(3)	1	2·703 (2)			
Cu(3)	1	2·819 (2)	Cu(4)–Hf(1)	2	2·866 (1)
Cu(4)	1	2·866 (1)	Hf(1)	2	2·875 (2)
Cu(4)	1	2·875 (2)	Hf(2)	1	3·036 (3)
Cu(5)	1	2·811 (1)	Cu(1)	2	2·798 (3)
Cu(6)	1	2·895 (1)	Cu(2)	2	2·492 (2)
Cu(6)	1	2·909 (2)	Cu(5)	1	2·403 (4)
			Cu(5)	1	2·558 (4)
Hf(2)–Cu(1)	2	2·941 (2)	Cu(6)	1	2·430 (4)
Cu(1)	2	2·793 (2)			
Cu(2)	2	2·760 (2)	Cu(5)–Hf(1)	2	2·811 (1)
Cu(2)	2	2·789 (2)	Hf(2)	1	2·886 (3)
Cu(3)	1	2·633 (3)	Cu(1)	2	2·528 (3)
Cu(3)	1	3·112 (3)	Cu(1)	2	2·617 (3)
Cu(4)	1	3·036 (3)	Cu(2)	2	2·527 (2)
Cu(5)	1	2·886 (3)	Cu(4)	1	2·403 (4)
Cu(6)	1	2·906 (3)	Cu(4)	1	2·558 (4)
			Cu(6)	1	2·501 (3)
Cu(1)–Hf(1)	1	2·666 (2)			
Hf(1)	1	2·697 (2)	Cu(6)–Hf(1)	2	2·895 (1)
Hf(2)	1	2·793 (2)	Hf(1)	2	2·909 (2)
Hf(2)	1	2·941 (2)	Hf(2)	1	2·906 (3)
Cu(1)	1	2·817 (4)	Cu(1)	2	2·552 (3)
Cu(2)	1	2·658 (3)	Cu(2)	2	2·492 (2)
Cu(2)	1	2·668 (2)	Cu(3)	1	2·438 (3)
Cu(2)	1	2·895 (2)	Cu(4)	1	2·430 (4)
Cu(3)	1	2·964 (2)	Cu(5)	1	2·501 (3)
Cu(4)	1	2·798 (3)			
Cu(5)	1	2·617 (3)			
Cu(5)	1	2·528 (3)			
Cu(6)	1	2·552 (3)			
Cu(2)–Hf(1)	1	2·910 (2)			
Hf(1)	1	2·886 (2)			
Hf(1)	1	3·014 (2)			
Hf(2)	1	2·760 (2)			
Hf(2)	1	2·789 (2)			
Cu(1)	1	2·658 (2)			
Cu(1)	1	2·668 (2)			
Cu(1)	1	2·895 (2)			
Cu(3)	1	2·506 (2)			
Cu(4)	1	2·492 (2)			
Cu(5)	1	2·527 (2)			
Cu(6)	1	2·492 (2)			

sequence repeated after four layers. The atoms in the fourfold positions are situated on the mirror planes at $y=\frac{1}{4}$ and $y=\frac{3}{4}$, forming planar layers. The atoms in the eightfold positions, with approximate y coordinates of 0 and $\frac{1}{2}$, form puckered layers sandwiched between the mirror planes. A projection on (010) of the puckered layer near $y=0$ together with the planar layer at $y=\frac{1}{4}$ is shown in Fig. 1. Fig. 1 also shows a projection of the puckered layer near $y=\frac{1}{2}$ together with the planar layer at $y=\frac{3}{4}$.

The various interatomic distances in Hf_3Cu_8 are given in Table 4. Only distances shorter than 3·5 Å are included.

With the exception of the eightfold Hf(1) and Cu(1) atoms, which have, respectively, one Hf(1) and one Cu(1) contact across a mirror plane, the atoms in one layer have near neighbours only within the same layer and in the two adjacent layers. The Hf(1) atoms are coordinated to two other Hf(1) at 3·083 and 3·175 Å and to 12 Cu atoms, while the Hf(2) atoms have 13 Cu neighbours. The Hf–Cu distances are between 2·633 and 3·112 Å. The coordination numbers for the six non-equivalent types of Cu atoms vary from 11 to 13, with Cu–Cu distances from 2·403 to 2·964 Å.

The coordination of the atoms in Hf_3Cu_8 may be compared with that in $Hf_{14}Cu_{51}$ (Gabathuler, White & Parthé, 1975). In $Hf_{14}Cu_{51}$ there is only one Hf–Hf contact at a distance of 3·106 Å. The coordination numbers for the three non-equivalent Hf atoms are 16, 14 and 14, while the coordination numbers for the Cu atoms vary from 11 to 13. The Hf–Cu and Cu–Cu distances range from 2·660 to 2·974 Å, and from 2·413 to 3·269 Å respectively. Apart from the larger number of neighbours to the Hf atoms in $Hf_{14}Cu_{51}$, the coordination numbers and interatomic

distances are evidently quite similar in $Hf_{14}Cu_{51}$ and Hf_3Cu_8.

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